



Ho³⁺ activated Ca_{0.5}Y_{1.90-x}O₃ green-emitting nanophosphors for solid state lightening: Synthesis, characterization and photoluminescence properties

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ABSTRACT

This paper presents a series of Ho³⁺ activated Ca_{0.5}Y_{1.90-x}O₃ green-emitting nanophosphors using the facile solution combustion method. X-ray diffraction (XRD) and Transmission electron microscope (TEM) confirm the phase purity, structural morphology with cubic phase, and irregular shape particle with the average particle size of 21 nm. Diffuse reflectance spectra of the recorded and band gap estimated by Tauc's equation which is 5.78 eV, 4.90 eV for direct and indirect band gaps for 2 mol% of Ho³⁺, respectively. Under the excitation of 448 nm wavelength (⁵I₈ → ³G₆), the doped samples display characteristic emission of Ho³⁺ ions correspond to ⁵S₂ → ⁵I₈ (at 551 nm) transition, which fall in a strong green region. The optimal concentration for Ho³⁺ doping is found as 2 mol%, after this quenching of photoluminescence (PL) intensity occurred and this is accredited due to dipole-dipole interaction. The proposed nanophosphor also displays the good thermal stability with activation energy of 0.23 eV. The nanophosphor confirms the excellent color purity (98%), CRI (98.36%), and CCT value (5724 K). With this, nanophosphor also exhibits good quantum efficiency of 81.1% under the 448 nm excitation wavelength. These results direct that the Ho³⁺ activated Ca_{0.5}Y_{1.90-x}O₃ nanophosphor may have potential applicability for green-emitting LEDs, display device and optoelectronic applications.

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1. Introduction

Nowadays, there is huge demand for highly efficient LEDs and display devices, which have less power consumption properties. Rare earth doped inorganic materials are considered superior as they can provide narrow emission with UV excitation. This property makes it available for various fields like in lamp industry, color display, medical imaging, laser and security devices etc. [1–3]. For the lamp industries the rare earth doped phosphor material gained much attention for fabrication of the light emitting diodes (LEDs). Hence, for realizing an efficient LED, phosphor material must have good luminescence efficiency, narrow emission and high color rendering index which define it an efficient material for optoelectronic applications [4–6]. Generally, rare earth doped material has all these properties and is used as activator ions, because it has forbidden 4f–4f transitions for the same parity and does not follow the Laporte selection rules. However, doping with the non-

centrosymmetric state with opposite parity ligands, leads to parity allowed transitions. These result in the emission in a narrow band, as 4f electrons of rare-earth ions are shielded by large 5s²5p⁶ radial subshells [6,7].

In various activator rare earth ions, holmium (Ho³⁺) ions are an important activator ion as it has a broad PL spectrum from visible to IR range with strong green emission [2,6,8]. For the Ho³⁺ ions green emission dominant due to ⁵F₄; ⁵S₂ → ⁵I₈ transitions over the blue and red emission, this is also mentioned in the Singh et al. [5] and Atabaev et al. [8]. The efficient green emission of Ho³⁺ ions can be applicable in various fields like solid state lighting devices, field emission display, security printing and for biological and sensing applications [3,4,9]. To obtain an efficient phosphor it is important to select a proper host and Y₂O₃ are the most suitable host for doping with rare earth. As the Y₂O₃ has similar chemical and physical properties with various rare earth ions such as large transparency range and also have similar ionic radii, high melting point minimum and thermal expansion coefficient [8]. Rare earth doped Y₂O₃ also has a wide range of applications [10,11]. Further, to improve luminescence qualities co-doping can be done, which can be used as a sensitizer by effectively transferring en-

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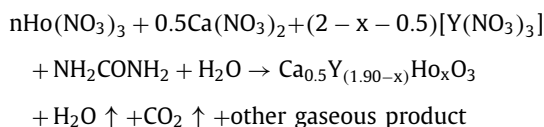
ergy to the activator rare earth ions (Ho^{3+}). Sometimes co-doping generates defects which improve the luminescence property of the material with high quantum efficiency. There is various work which shows the co-doping enhances the luminance intensity by increasing transition probability [9,12]. The monovalent ions (Li^+ , Na^+ , K^+) and divalent ions (Ca^{2+} , Sr^{2+} , Mg^{2+}) are widely used as a common co-dopant material to improve the performance of phosphor material by enhancing PL intensity [9,10]. The divalent ions (Ca^{2+} , Sr^{2+} , Mg^{2+}) and rare-earth ions have different valence states, which affect the lattice structure, creates distortion in the crystal field of rare earth ions and improve the optical properties of the phosphor material [13]. Kumari, Manam reported the improvement of PL emission intensity of $\text{YVO}_4:\text{Eu}^{3+}$ after doping with alkaline earth metal ion [12]. Shi, et al. reported the effect of doping of divalent ions with $\text{Eu}^{3+}:\text{Y}_2\text{O}_3$ [9]. However, doping with divalent ions with Ho^{3+} is less focused and studied and no detailed analysis with Ho^{3+} doped $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3$ nanophosphor has been studied so far.

The present work reports the effect of doping concentration of Ho^{3+} ions on the structural and optical properties of $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:\text{xHo}^{3+}$ nanophosphor. The samples of $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:\text{xHo}^{3+}$ nanophosphor is synthesis by solution combustion method using urea as a fuel, as best of our knowledge this combination of $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:\text{xHo}^{3+}$ phosphor material has not been investigated and reported yet. To study phase purity, structural and morphological characterization were carried out by X-ray diffraction (XRD) and Transmission electron microscope (TEM), respectively. Purity of the sample has been tested by FTIR. The optical properties are investigated through photoluminescence spectra and diffuse reflectance spectra (DRS). The optical band gap of the phosphor materials is also calculated from DRS. The decay lifetime corresponding to the $^5\text{S}_2$ state of different concentrations of Ho^{3+} was also studied and described. The photometric properties and color purity has been also studied using a chromaticity diagram. The energy transfer mechanism with an energy level diagram is also discussed.

2. Experimental method

2.1. Materials and synthesis

The starting material yttrium oxide (Y_2O_3), calcium oxide (CaO), and holmium oxide (Ho_2O_3), was purchased from Sigma-Aldrich. Urea (NH_2CONH_2) and nitric acid (HNO_3) was perched from Loba Chemie all are analytical and used without further purifications. A series of $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:\text{xHo}^{3+}$ ($x = 0.05, 1, 1.5, 2, 2.5, 3$ mole%) nanophosphor was synthesised by solution combustion method according to the solution:



For the synthesis, stoichiometric amounts of Y_2O_3 , Ho_2O_3 , and CaO reagents are mixed separately with the 2 ml HNO_3 and mixed separately in beakers until a transparent solution of it. Next, a separate solution of urea (used as fuel) with 10 ml deionised water (in 2:1 urea to metal ion ratio) was formed. The nitrate solution of all the reagents is mixed with solution of urea in a beaker followed by continuous starting of 30 min. Further, the obtained transparent mixture was kept in the furnish at 600°C for 1 hour. Finally, we get a white powder cooled to room temperature gridded and washed twice with ethanol followed by calcination at 900°C for 8 h and used for further characterizations.

2.2. Characterizations

X-ray diffraction of the synthesised sample was recorded through (XRD-Rigaku Mini Flax 600 diffractometer) with $\text{CuK}\alpha$ radiation ($\lambda = 1.54,046 \text{ \AA}$) for the 2θ range 10° – 80° at scan rate $0.02^\circ/\text{min}$. The BRUKER-ALPHA II FT-IR spectrometer was used to record Fourier transform infrared (FTIR). The morphological and microstructures characterization of the synthesised material was estimated through transmission electron microscope (TEM) TECHNICAL G² 20, with 200 kV accelerating voltage. Diffused reactance spectra were recorded with the Perkin Elmer Lambda 750 UV-vis NIR (ultraviolet-visible near-infrared) spectrometer. The Photoluminescence excitation and emission spectra of samples were studied by using the Perkin Elmer Photoluminescence spectrophotometer. To test the thermal stability the phosphor materials were heated outside and PL was measured. Finely, the Fluorolog-3 (Horiba Jobin Yvon) spectrofluorometer excited with PMT attached 25 W pulsed Xenon lamp in phosphorescence mode used for recording lifetime decay curves.

3. Result and discussion

3.1. X-ray diffraction characterization

The XRD profiles of $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:\text{xHo}^{3+}$ ($x = 0.05, 1, 1.5, 2, 2.5, 3$ mole%) samples were measured for $2\theta = 10^\circ$ – 80° range and presented in Fig. 1(a). Synthesized samples display high intensity and sharp peaks and refer to crystalline nature and are matched with the standard (JCPDS: 83–0928) data of Y_2O_3 . All the phosphors have similar exhibit similar XRD pattern with the most intense XRD peaks are $\sim 29.2^\circ, 33.8^\circ, 48.4^\circ, 57.6^\circ$ belongs to diffraction plane (222), (400), (440) and (622) plane, respectively. There are other peaks with low intensity that are also present and well matched with the JCPDS: 83–0928. This revealed samples are cubic structure with the $\text{Ia}\bar{3}$ (206) Space group. It has been observed that there is a small shift in XRD peaks (222) towards the higher angle side after doping with xHo^{3+} ($x = 0.05, 1, 1.5, 2, 2.5, 3$ mol%) (Fig. 1(b)). This shift in peak attributed to mismatch in radii of Y^{3+} (0.90 Å) and Ca^{2+} (1.1 Å) and Ho^{3+} (0.901 Å) during doping [14]. In the host ($\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3$) the ionic radii of Ho^{3+} is similar with that of Y^{3+} hence doping with Ho^{3+} most likely replaces Y^{3+} as the dopant has tendency to substitute similar ionic radii. This replacement of Ho^{3+} into the Y^{3+} sites of $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3$ was also favoured by charge balancing. As the doping concentration of doped ions (Ca^{2+} , Ho^{2+}) are low, impurity peaks of other phases have been not found, but the intensity of the XRD peaks decreases with increasing the concentration of Ho^{3+} .

Further, the crystallite size of $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:\text{xHo}^{3+}$ phosphor has been estimated with Debye–Scherrer (D-S) formula and Williamson–Hall (W-H) relation given as [12,15]:

$$D = \frac{0.89\lambda}{\beta \cos(\theta)} \quad (1)$$

$$\beta \cos(\theta) = \frac{K\lambda}{D} + 4\epsilon \sin(\theta) \quad (2)$$

where, λ is wavelength of x-ray (0.15,406 nm), θ is Bragg's angle of diffraction, β is the full width at half maxima, and ϵ is the strain present in the sample. The value of average crystallite size and strain present in the samples were calculated from the intercept and slope of the W-H plot given in Fig. 1(c). The particle size of the Debye–Scherrer method is less compared to the W-H method. It might be due to the W-H method including lattice strain also during particle size calculation [11]. It has been observed that lattice strain increases with doping, this might be

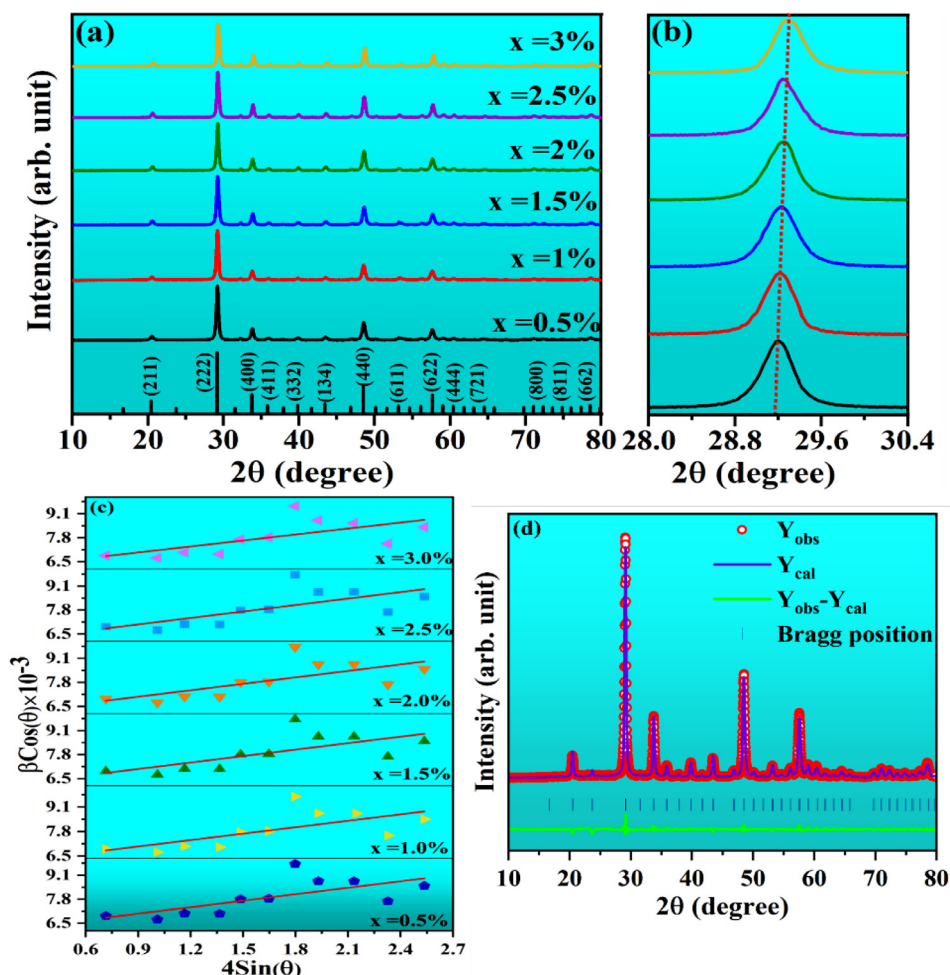


Fig. 1. (a) Powder XRD patterns and (b) magnified pattern of XRD for 28–30.4° (c) Hall-Williamson plot of $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:\text{xHo}^{3+}$ ($x = 0.05, 1, 1.5, 2, 2.5, 3$ mole%) phosphors (d) Rietveld refinement of $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:\text{xHo}^{3+}$ (2 mol%) nanophosphor.

Table 1
Structural parameters for $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:\text{xHo}^{3+}$ with different concentration of Ho^{3+} .

Nanophosphor $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:\text{xHo}^{3+}$ (for X mole%)	Crystallite size D_{hkl} (nm)		microstrain $\varepsilon \times 10^{-3}$	Dislocation density (nm^{-2}) $\delta \times 10^{-4}$
	D-S formula	W-H plot		
	$\text{Ho}^{3+}=0.5$	20.4	22.9	1.08
$X = 1$	20.9	23.1	1.13	18.6
$X = 1.5$	21.7	23.2	1.17	18.5
$X = 2$	22.9	23.2	1.17	18.5
$X = 2.5$	23.1	23.2	1.17	18.5
$X = 3$	23.8	24.2	1.18	17.0

due to doping with $\text{Ca}^{2+}/\text{Ho}^{3+}$ in Y_2O_3 . With the average crystallite size calculated from Eq. (2) other structural parameter like microstrain ($\varepsilon = \frac{\beta \cos(\theta)}{4}$) and dislocation ($\delta = \frac{1}{D^2}$) [12] has been also estimated and values are given in Table 1. These results also show that the doping ion creates a little distortion in the Y_2O_3 crystal field but does not affect the cubic phase of it.

Furthermore, the effect of doping of $\text{Ca}^{2+}/\text{Ho}^{3+}$ with Y_2O_3 on the lattice constants and cell volume has been calculated from XRD data through Rietveld refinement using FullProf software for 2 mol% doping of Ho^{3+} ions (Fig. 1(d)). The Goodness of Fit (GOF) ($\text{GOF} = \chi^2 = (R_p/R_w)^2$) was found to be 1.74, this confirms good fitting between theoretical and experimental plots. Refinement also conforms the cubic phase with space group $\text{Ia}\bar{3}$ (206). Moreover,

it has been estimated that doping increases lattice constant from 10.61 Å (JCPDS: 83–0928) to 10.62 Å and volume 1194 Å³ to 1197.1 Å³ for 2 mol% doping. This might be due to doping with comparatively bigger ions ($\text{Ca}^{2+}/\text{Ho}^{3+}$) than Y^{3+} [15].

3.2. Morphological characterization

The surface morphology and particle size distribution of Ho^{3+} doped nanophosphor sample was investigated with TEM. Fig. 2 depict the TEM image of 2 mol% Ho^{3+} doped $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:\text{xHo}^{3+}$ nanophosphor. It can be seen from the TEM (Fig. 2(a)) image that particles are in the nanometre range. These particles have small average crystallites size ~21 nm (Fig. 2(b)) with irregular in shape and slightly agglomerated. Here, in the present study, urea has been used as a fuel which may expedite the nucleation of particles, thereby resulting in the agglomerates after combustion synthesis [16,17]. It has been also observed that average particle size values are consistent with the estimated values from the XRD pattern using scherrer's equation. The HRTEM image given in Fig. 2(c) show that the distance between the fringes is 0.305 nm, this can be allocated to (222) plane of crystal [8]. The SAES pattern (Fig. 2(d)) also shows the crystalline nature of $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:\text{xHo}^{3+}$ and index with (222), (400), (440), (622) plane [18,19]. In SAED pattern several diffraction spots, these spots are ordered in ring form which resemble to high nano crystallinity with polycrystalline nature of sample.

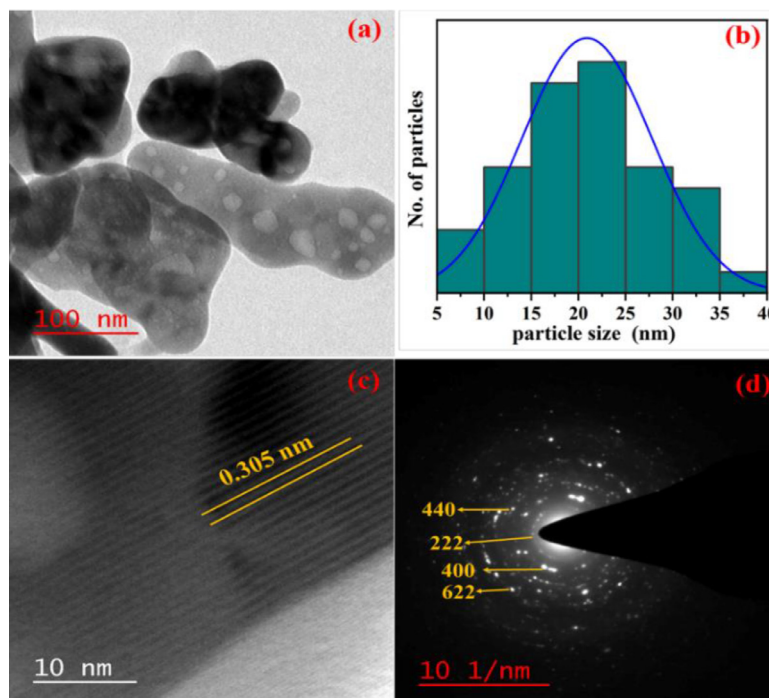


Fig. 2. (a) TEM image (b) particle distributions histogram (c) lattice fringes (d) SAED analysis of $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:\text{xHo}^{3+}$ (2 mole%) nanophosphor.

3.3. Fourier transform- infrared (FT-IR)

FT-IR spectra are generally used to identify organic and inorganic functional groups attached with the prepared samples for the range 500 cm^{-1} to 4000 cm^{-1} given in the Fig. 3. Which clearly confirms the absorption band of Y-O stretching near 558 cm^{-1} [19]. The band near 855 cm^{-1} confirms the C-O vibration of CO_3^{2-} and absorption near 1433 cm^{-1} represents symmetric and asymmetric vibrations of C-O group. The presence of this group may be due to absorption due to atmosphere or residue carbon remaining during synthesis [20]. With this, the band centered at $3400\text{--}3600\text{ cm}^{-1}$ and 1500 cm^{-1} display stretching and bending vibrations of -OH group [9]. It can be seen that no other residual peaks are present hence, this indicates pure phase formation of samples.

3.4. Diffuse reflectance spectra (DRS) characterization

The diffuse reflectance spectra of $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:\text{xHo}^{3+}$ is shown in Fig. 3(a), for the range 200-700 nm. The dominant peak at $\sim 220\text{ nm}$ is accredited to host absorption, with this there are several less intense peaks at 360 nm, 448 nm, 463, 540, and 648 nm are corresponding to ${}^3\text{H}_6$, ${}^5\text{G}_6$, ${}^5\text{F}_3+{}^3\text{K}_8$, ${}^5\text{F}_4+{}^5\text{F}_3$, ${}^5\text{S}_2$, and ${}^5\text{F}_5$ transition of Ho^{3+} ions. These peaks are arising due to 4f-4f electronic transitions from ground state (${}^5\text{I}_8$) to upper excited states of Ho^{3+} ions [5,8]. This result confirms the successful doping of Ho^{3+} in $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3$ host. Further, the optical bandgap of $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:\text{xHo}^{3+}$ was calculated with the Tauc's equation from the Tauc plot given in Fig. 3 (b & c) [6,11].

$$\alpha h\nu = C(h\nu - E_g)^n \quad (3)$$

where, C is proportionality constant and $h\nu$ is photon energy, α is the absorption coefficient, E_g is the energy of optical band gap. The value of 'n' is different for different transitions as $1/2$ for direct allowed, $3/2$ for direct forbidden, 2 indirect allowed and 3 for indirect forbidden transition, respectively. Here, we consider direct and indirect allowed transitions so in this case the value of 'n' is $1/2$ and 2 [18]. The optical band gap E_g was calculated by plotting $h\nu$ versus $(\alpha h\nu)^n$ by extrapolating the linear region of the plot up to $\alpha = 0$.

The estimated values for energy band gap of direct allowed transition are 5.74, 5.72, 5.71, 5.78, 5.69, 5.07 eV and for indirect allowed transition are 5.0, 5.02, 5.00, 4.90, 4.91, 4.99 eV for different concentration of Ho^{3+} ions $x = 0.5, 1, 1.5, 2, 2.5, 3$ mol%, respectively. This high value indicates host material has excellent PL properties [6]. It has been observed that there is slight variation of energy band gap for direct and indirect transitions with increase of Ho^{3+} ion concentration; this variation in band gap is also mentioned by Raman et al. [4].

3.5. Photoluminescence (PL) analysis

The photoluminescence excitation spectra of $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:\text{xHo}^{3+}$ with different concentration of Ho^{3+} ($x = 0.05, 1, 1.5, 2, 2.5, 3$ mole%) were given in Fig. 4(a), which was recorded for the range 320-500 nm. The spectra are monitored for the emission wavelength $\lambda_{em} = 551\text{ nm}$, and contains several peaks at 336 nm (${}^5\text{I}_8 \rightarrow {}^3\text{F}_4$), 345 nm (${}^5\text{I}_8 \rightarrow {}^3\text{L}_9$), 361 nm (${}^5\text{I}_8 \rightarrow {}^3\text{H}_5$), 374 nm (${}^5\text{I}_8 \rightarrow {}^5\text{G}_4$), 448 nm (${}^5\text{I}_8 \rightarrow {}^5\text{G}_6$), 459 nm (${}^5\text{I}_8 \rightarrow {}^3\text{K}_8$), 460 nm (${}^5\text{I}_8 \rightarrow {}^5\text{F}_2$) and 484 nm (${}^5\text{I}_8 \rightarrow {}^5\text{F}_3$) correspond to the transitions. These excitation peaks are arisen due to f-f transitions from ground state of Ho^{3+} to various excited states [4] similar, type of transition was also mentioned and reported in Singh et al. [6]. Amongst these transitions, the 448 nm (${}^5\text{I}_8 \rightarrow {}^3\text{G}_6$) is most instance, hence chosen for recording the excitation of the sample.

The PL emission spectra of $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:\text{xHo}^{3+}$ recorded for the excitation wavelength at $\lambda_{ex} = 448\text{ nm}$ for the 500-700 nm wavelength has been presented in Fig. 4(b). The emission peaks consist three emissions at 539 nm (${}^5\text{F}_4 \rightarrow {}^5\text{I}_8$) and 551 nm (${}^5\text{S}_2 \rightarrow {}^5\text{I}_8$) lie in strong green region and a weak peak at 670 nm (${}^5\text{F}_5 \rightarrow {}^5\text{I}_8$) in red region which are characteristic peaks of Ho^{3+} and confirms the presence of Ho^{3+} ion in the synthesized samples [21]. It can be seen that (${}^5\text{S}_2 \rightarrow {}^5\text{I}_8$) transitions at 551 nm is dominant over the other transitions in $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:\text{xHo}^{3+}$ nanophosphor. Hence it can be said that in all the transitions, (${}^5\text{S}_2 \rightarrow {}^5\text{I}_8$) transition is most instance which lie in green region of visible light spectra. Further the effect of doping concentration on the PL emission intensity has been observed as given in Fig. 5(a).

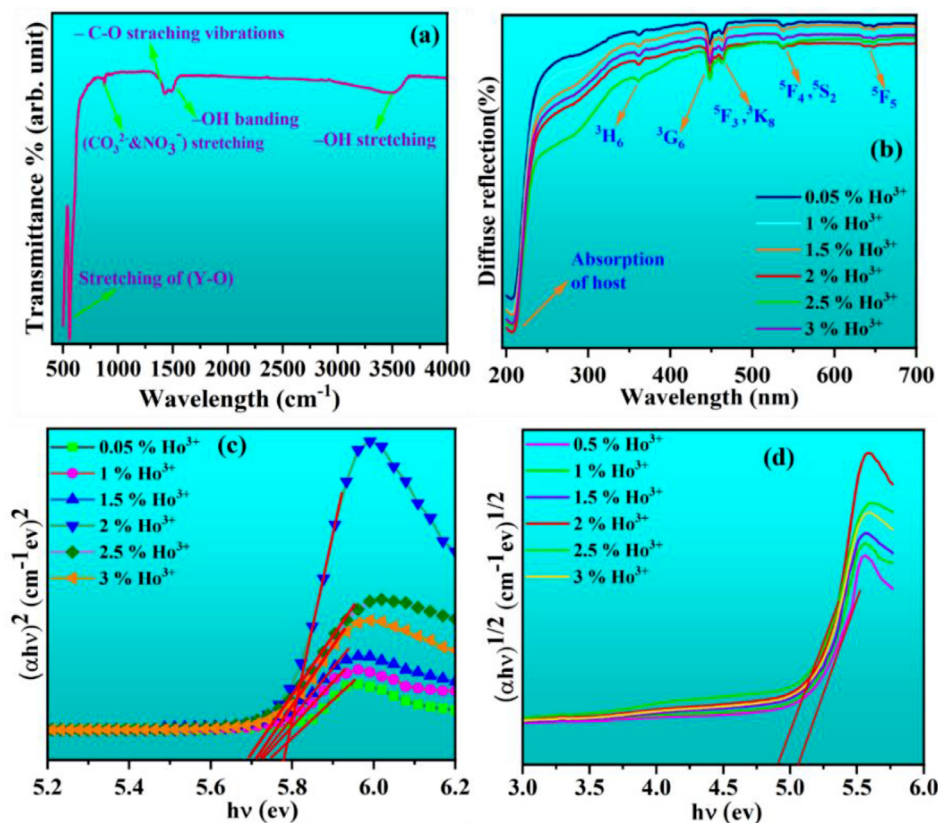


Fig. 3. (a) FT-IR spectrum of $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:\text{xHo}^{3+}$ (2 mole%) nanophosphor (b) UV-Vis diffuse reflectance spectra (c) Tauc plot of $(\alpha h\nu)^2$ versus $h\nu$ and (d) $(\alpha h\nu)^{1/2}$ versus $h\nu$ of $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:\text{xHo}^{3+}$ for different concentrations of Ho^{3+} .

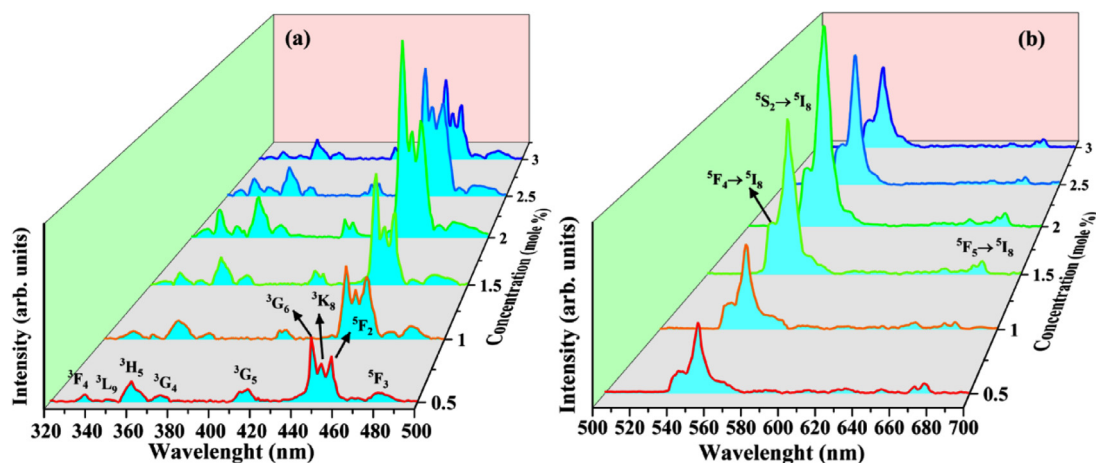


Fig. 4. (a) Excitation spectra of under $\lambda_{em} = 551$ nm (b) emission spectra under $\lambda_{ex} = 448$ nm of $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:\text{xHo}^{3+}$ ($x = 0.5, 1, 1.5, 2, 2.5, 3$ mole%) phosphors.

It has been seen that intensity increases up to 2 mol% and further decreases with increasing concentration of Ho^{3+} , and this owing to concentration quenching effect [5]. The phenomena of concentration quenching can be elucidated in two ways; one of them is due to presence of defects and impurity ions in the crystal lattice and causes energy transfer through the lattice vibration and leads to concentration quenching. In the second process, excited state energy transfer arises through cross relaxation/nonradiative transitions and this phenomenon is strongly influenced by doped ion concentration [4, 6]. This energy transfer mechanism through cross-relaxation/nonradiative transitions process may be understood by the energy level diagram given in Fig. 6. The non-radiative energy transfer is also one of the reasons for the concentration quenching.

This might be due to when Ho^{3+} ion concentration is increased then they become close enough to each other or forms agglomerated structure, which leads to effective energy transfer through energy transfer and non-radiative cross-relaxation between Ho^{3+} ions and reduces the PL intensity efficiently [4]. Hence, the 2 mol% is considered as an optimal concentration of Ho^{3+} ion in $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:\text{xHo}^{3+}$ nanophosphor.

3.6. Energy transfer studies of $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:\text{xHo}^{3+}$ nanophosphor

Further, in order to understand the energy transfer mechanism, the critical distance between the $\text{Ho}^{3+}-\text{Ho}^{3+}$ ions can be calculated

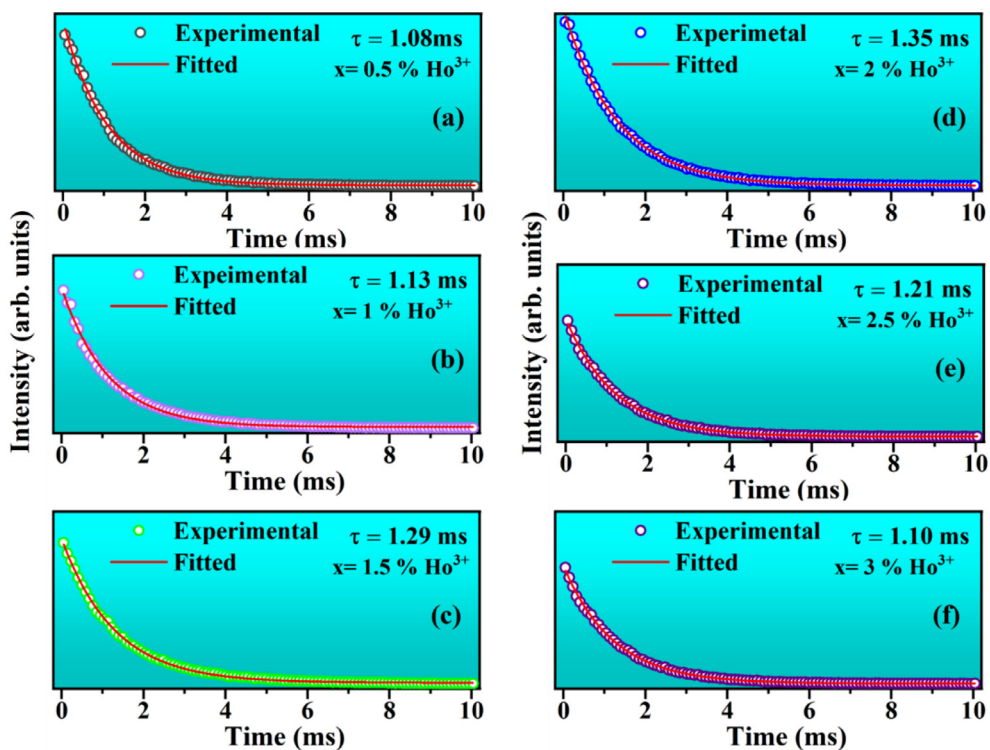


Fig. 7. Luminescence decay curve of $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:x\text{Ho}^{3+}$ phosphors with different concentration of Ho^{3+} ions.

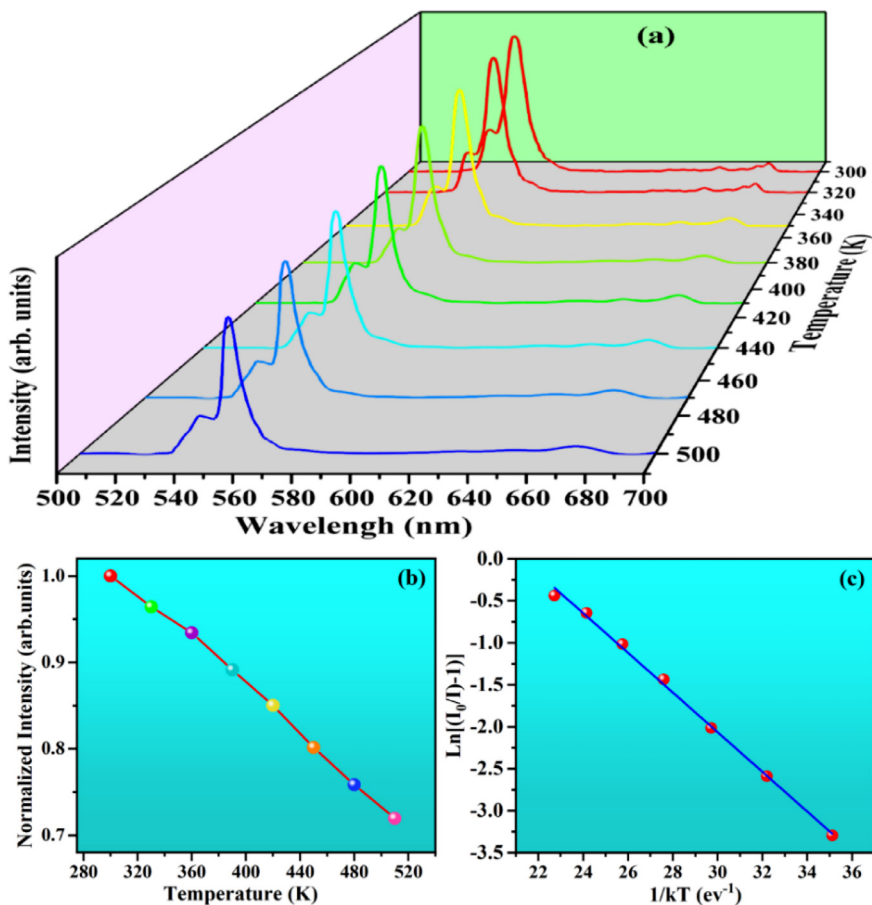


Fig. 8. (a) & (b) PL emission spectra as a function of temperature for $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:2.0\text{Ho}^{3+}$ nanophosphor (c) Arrhenius plot of $\ln[(I_0/I)-1]$ versus $1/kT$ for thermal quenching nature of $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:2.0\text{Ho}^{3+}$ sample.

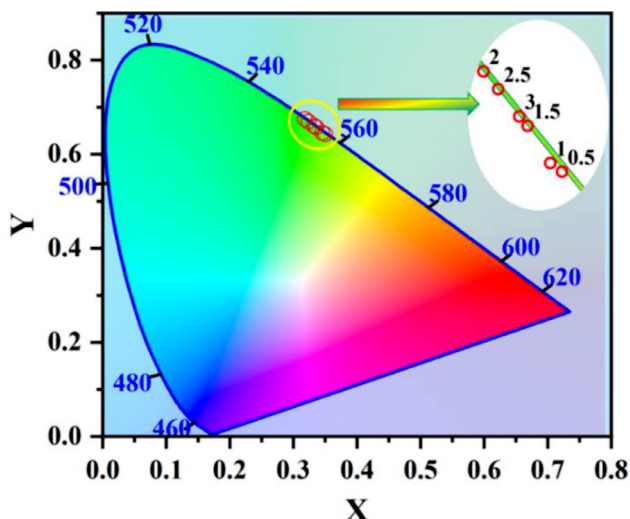


Fig. 9. The CIE 1931 chromaticity diagram of $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:\text{xHo}^{3+}$ ($x = 0.05, 1, 1.5, 2, 2.5, 3$ mole%) phosphors.

the phosphor Arrhenius Eq. (6) has been employed to calculate the activation energy [27,30]:

$$\ln\left(\frac{I_0}{I} - 1\right) = \ln A - \frac{\Delta E}{kT} \quad (6)$$

where, I_0 and I initial emission intensity and intensity at different temperatures, k is Boltzmann constant, A is constant and ΔE is ascribed as activation energy. According to Arrhenius equation $\ln[(I_0/I)-1]$ versus $1/kT$ plot (Fig. 8(c)) has been plotted, and fitted linearly with the slope -0.237 eV. The slope of this plot gives activation energy, therefore ΔE the thermal quenching is estimated as 0.237 eV. Hence the $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:2.0\text{Ho}^{3+}$ nanophosphor has good thermal stability and can be used for the LEDs fabrication.

3.9. Photometric analysis

To recognize the purity in emitted color of the synthesised nanophosphor, the color coordinate was investigated with the CIE 1931 (Commission International de l'Éclairage). The CIE coordinated with different concentrations of Ho^{3+} was estimated with the emission data presented in Fig. 9. The CIE coordinates for different concentrations of Ho^{3+} given in Table 2. It can be seen that color coordinates lie in the bright green spectral region which well match with the PL results. Hence, the $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:\text{xHo}^{3+}$ nanophosphor might be used for the solid-state lighting device to give emission in the green region. Further, the percentage color purity exhibited by the $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:\text{xHo}^{3+}$ can be estimated through the formula (7) [6,31]:

$$\text{Color purity} = \frac{\sqrt{(x-x_w)^2 + (y-y_w)^2}}{\sqrt{(x-x_d)^2 + (y-y_d)^2}} \times 100\% \quad (7)$$

Table 2

The CIE coordinates, CCT coordinate, CCT values, % color purity and CIR value of $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:\text{xHo}^{3+}$ nanophosphors with distinct concentration of Ho^{3+} .

$\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:\text{xHo}^{3+}$ Nanophosphor (at. mole%)	CIE coordinates (x, y)		CCT value (K)	Color purity (%)	CRI
0.5	0.349	0.643	5260	90	98.50
1	0.344	0.645	5330	91	98.49
1.5	0.335	0.657	5467	94	98.44
2	0.317	0.675	5724	98	98.36
2.5	0.321	0.669	5639	97	98.39
3	0.332	0.660	5516	95	98.42

where, (x,y) present the color coordinates of sample point, (x_w,y_w) denote the coordinate of the illuminant point ($x_w = 0.3101, y_w = 0.3162$) and (x_d,y_d) represents the coordinated of the dominant wavelength ($x_d = 0.318, y_d = 0.676$). The calculated values of color purity values are given in Table 2, these values obtained indicate high purity of green emission in nanophosphor.

Furthermore, the correlation color temperature (CCT), the values have been calculated as CCT present the quality of light source and it is used to show cool and warm nature of light. The CCT values can be estimated by McCamy's empirical formula (8) [4,32]:

$$T = -499n^3 + 3525n^2 - 6823.3n + 5520 \quad (8)$$

where, $n = \frac{x-0.332}{y-0.186}$ and x, y is CIE coordinated. The obtained CCT values of $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:\text{xHo}^{3+}$ nanophosphors are given in Table 2. All the CCT values are greater than 3000, which show these lies in the cool light region and can be said that it is a stable material with prodigious use for cool LEDs and display applications [33,34]. Color purity of the prepared phosphor samples was also estimated and tabulated in Table 2. The samples display high color purity and the green color purity reached maximum 98% for the 2 mol% doped $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:\text{xHo}^{3+}$ nanophosphors.

3.10. Quantum efficiency (QE)

With the color purity, chromatic properties, thermal stability quantum efficiency is one of the important parameters of the synthesised nanophosphor to reorganize their applicability for LEDs applications. The quantum efficiency (η) of the optimized $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:\text{Ho}^{3+}$ (2 mole%) has been estimated by following method (9) [35,36]:

$$\text{QE}(\eta) = \frac{\text{number of photons emitted } (\varepsilon)}{\text{number of photons absorbed } (\alpha)} = \frac{E_c - E_a}{L_a - L_c} \quad (9)$$

where, E_c is the integrated luminescence of the phosphor caused by direct excitation, E_a is the integrated luminescence from the empty integrating sphere (without sample), L_a is the integrated excitation profile from the empty integrating sphere, and L_c is the integrated excitation profile when the sample is directly excited by the incident beam. The QE of the sample with $\lambda_{ex} = 448$ nm excitation wavelength was evaluated by integrated emission count for 500-700 nm range. The value has found to be $\sim 81.1\%$. It is evident that the estimated value of the QE in this case is higher and comparable with the other green emitting phosphor, such as in case of Ho^{3+} doped $\text{Ba}_{0.65}\text{Sr}_{0.35}\text{TiO}_3$ ($\sim 57\%$) [37], $\text{YAlO}_3:\text{Ho}^{3+}$ ($\sim 61\%$) [36], and $\text{Ho}^{3+}:\text{LaF}_3$ ($\sim 55.8\%$) [38]. This result indicates that the $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:\text{Ho}^{3+}$ (2 mole%) nanophosphors are promising for the LEDs.

4. Conclusion

Our study reports the successful synthesis of $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:\text{xHo}^{3+}$ with different concentrations of Ho^{3+} ions ($x = 0.5, 1, 1.5, 2, 2.5, 3$ mole%) nanophosphor through solution combustion method. XRD reveal the cubic phase with the $\text{Ia}\bar{3}$ (206) Space group and Transmission electron microscope (TEM) confirms the irregular shape particle in nanometre range ~ 21 nm, respectively. The band gap estimated by Tauc's equation is 5.78 eV, 4.90 eV for direct and indirect band gaps for 2 mol% of Ho^{3+} , respectively. The PL excitation spectra show numerous peaks corresponding to f-f transitions of Ho^{3+} for 551 nm emission. The PL emission spectra shows strong green correspond to $^5\text{S}_2 \rightarrow ^5\text{I}_8$ (551 nm) transition of Ho^{3+} ions. The critical concentration of Ho^{3+} in $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:\text{xHo}^{3+}$ was detected as 2 mol% above this concentration quenching effect has been observed. The

$\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:2.0\text{Ho}^{3+}$ green emitting phosphor shows excellent thermal stability with activation energy 0.27 eV. Photometric properties have been also obtained using chromaticity diagram for the optimal concentration (2 mol%) as CIE coordinates (0.317, 0.675), CCT (5724 K), color purity (98%), CRI (98.36%) and quantum efficiency (~81.1%). The various characterization studies manifested strong green emission (at 551 nm) of $\text{Ca}_{0.5}\text{Y}_{1.90-x}\text{O}_3:\text{xHo}^{3+}$ nanophosphor which can be used for developing green emitting LEDs and display devices.

Declaration of Competing Interest

We confirm that this work is original and has not been submitted/published elsewhere. The article is exclusively submitted to **"Journal of Molecular Structure"**. We have no conflict of interest associated with this publication and there has been no significant financial support for this work that would have influenced its outcome. As a corresponding Author, I confirm that the manuscript has been read and approved for submission by all the named authors.

CRediT authorship contribution statement

Arpita Dwivedi: Methodology, Conceptualization, Data curation, Visualization, Validation, Writing – original draft, Writing – review & editing. **Monika Srivastava:** Formal analysis, Validation. **S.K. Srivastava:** Conceptualization, Validation, Supervision, Visualization.

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