

# PREFACE

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Semiconducting high- $\kappa$  dielectric and ferroelectric materials have technological applications in optoelectronic devices for uses in optical communication, memory, displays, and coherent optical processing. In semiconducting materials, high- $\kappa$  dielectricity and ferroelectricity are a result of the delicate balance between a long-range dipole–dipole interaction and a short-range interaction. Slight distortion of electron cloud due to structural changes in dielectrics gives rise to dipole moments. The ferroelectric phase transitions are associated with a structural phase transition from the high-symmetry paraelectric phase to the low-symmetry ferroelectric phase. Therefore, new materials with a simple structure are not only preferable for understanding the microscopic origin of ferroelectricity but are also important for integrating into modern ferroelectric devices with the utilization of novel ferroelectric/piezoelectric/relaxor and multiferroic materials that do not contain environmental hazards such as Pb and Bi.

Also, there is an interest in bringing  $d^0$  and  $d^n$  ( $0 \leq n \leq 10$ ) transition metal atoms together in a simple oxide matrix to realize multiferroic properties. Further altering of the relative ionicity and bonding character in M-O bond by structural anisotropy or introducing different electronic state cations in a single lattice to result in entropy-driven relative net polarization in the lattice can also act as an effective way to develop high- $\kappa$  dielectric and relaxor-type ferroelectrics. For the development of high- $\kappa$  dielectrics and relaxor-type ferroelectrics, in this thesis, the role of oxygen vacancy, and doping of multiple cations to develop structural anisotropy and to alter the ionic strength or ionicity in M-O to develop net polarization in the lattice are explored.

**Chapter 1** presents the introduction of dielectricity and its application to modern electronics and a way forward to develop superior high- $\kappa$  dielectric and ferroelectrics and understand the origin of high- $\kappa$  dielectricity and ferroelectricity.

**Chapter 2** deals with synthesis schemes and characterization techniques utilized in the study and the use of impedance spectroscopy to measure the dielectric properties of the material.

**Chapter 3** presents the investigation of  $\text{Cu}^+$  ion substituted ZnO and the role of  $d^0$ - $d^{10}$  interaction in the wurtzite structure and the effect of relative ionicity in M-O (M = Zn and Cu) bond to result in effective net polarization in the lattice.  $\text{Cu}^+$  ion substitution on  $\text{Zn}^{2+}$  sites not only resulted in net polarization but also opened the path for strong polarization between O-2p and filled  $d^{10}$  Zn/Cu 4p orbital in tetrahedral coordination. Up to 8% of  $\text{Cu}^+$  ions were substituted in the ZnO lattice and the highest dielectric constant ( $\sim 6300$ ) was obtained at  $600^\circ\text{C}$  for  $\text{Zn}_{0.95}\text{Cu}_{0.05}\text{O}_{1-\delta}$  at 100 KHz frequency. The materials  $\text{Zn}_{0.95}\text{Cu}_{0.05}\text{O}_{1-\delta}$  also exhibit ferroelectricity at room temperature with remnant polarization  $P_r$  and  $V_c$  equal to  $9.60 \times 10^{-03} \mu\text{C}/\text{cm}^2$  and  $3.83 \times 10^{+02} \text{V}/\text{cm}$  respectively.

**Chapter 4** presents the investigation of dielectric properties of bulk pristine  $\text{Cu}^{2+}$  and  $\text{Li}^+$  co-doped ZnO.  $\text{Zn}_{0.92}\text{Cu}_{0.05}\text{Li}_{0.03}\text{O}_{1-\delta}$  and  $\text{Zn}_{0.9}\text{Cu}_{0.05}\text{Li}_{0.05}\text{O}_{1-\delta}$  ceramics samples were prepared by the modified sol-gel method. The complex impedance spectroscopic study of Cu and Li co-doped ZnO confirms the high dielectric constant and low dielectric loss at high temperatures and high frequency compared to ZnO.  $T_m$  was found to increase with increasing frequencies suggesting the relaxor nature of dielectricity in the sample. The bulk pristine  $\text{Zn}_{0.9}\text{Cu}_{0.05}\text{Li}_{0.05}\text{O}_{1-\delta}$  also exhibits ferroelectricity and paramagnetic properties at room temperature. Bulk pristine  $\text{Zn}_{0.9}\text{Cu}_{0.05}\text{Li}_{0.05}\text{O}_{1-\delta}$  also exhibits

ferroelectricity at room temperature with remnant polarization  $P_r$  and  $V_c$  equal to  $4.20 \times 10^{-02} \mu\text{C}/\text{cm}^2$  and  $4.1 \times 10^{+03} \text{ V}/\text{cm}$  at (30KV, 500 Hz) respectively. Bulk pristine  $\text{Zn}_{0.9}\text{Cu}_{0.05}\text{Li}_{0.05}\text{O}_{1-\delta}$  also shows the dilute paramagnetic behavior with a coercivity and saturation magnetization reaching 114 Oe and 7.43 memu/g at 5K.

**Chapter 5** presents the synthesis of single-phase Fe-doped ZnO and Fe/Li co-doped ZnO and the investigation of the dielectric behavior of the materials. The complex impedance spectroscopic study of Fe-doped and Fe/Li co-doped ZnO shows a high dielectric constant and low dielectric loss at high temperatures and high frequency compared to ZnO. The Fe and Fe/Li doping depresses the concentration of the intrinsic donor and impedes the conduction mechanism resulting in the dielectric constant ( $\epsilon_r'$ ) equivalent to 612 for  $\text{Zn}_{0.9}\text{Fe}_{0.1}\text{O}_{1+\delta}$  and 90000 for  $\text{Zn}_{0.8}\text{Li}_{0.1}\text{Fe}_{0.1}\text{O}$ . at 1000 Hz frequency at 400°C. Also with an increase in frequency, the dielectric constant and dielectric loss were found to decrease.

**Chapter 6** presents the investigation of single-phase  $\text{Ti}^{4+}$  ion substituted  $\text{CoNb}_2\text{O}_6$  in the form of  $\text{CoNb}_{2-x}\text{Ti}_x\text{O}_6$ .  $\text{Ti}^{4+}$  substitution in the Trirutile  $\text{CoNb}_2\text{O}_6$  lattice enhances the dielectric constant of the material. Similar to well-known PZT type high  $\kappa$  Ferroelectrics,  $\text{Ti}^{4+}$  ion substitution in novel Trirutile  $\text{CoNb}_2\text{O}_6$  lattice resulted in high  $\kappa$  dielectricity and overall turned Trirutile into a new family of ferroelectrics as smaller  $\text{Ti}^{4+}$  can vibrate from its mean position in Trirutile octahedral resulting in net polarization in the octahedral. The dielectric constant ( $\epsilon_r'$ ) for  $\text{CoNb}_2\text{O}_6$  was found to be 500,  $\text{CoNb}_{1.95}\text{Ti}_{0.05}\text{O}_6$  is 700 and  $\text{CoNb}_{1.9}\text{Ti}_{0.1}\text{O}_6$  is 14000 respectively at 100 Hz frequency at 200°C and then decreases, clearly shows relaxor type behavior. Samples also exhibit ferroelectric behavior with remnant polarization  $P_r$  and  $V_c$  at 50Hz frequency equal to

$0.05 \mu\text{C}/\text{cm}^2$  and  $8 \times 10^{+03} \text{V}/\text{cm}$  for  $\text{CoNb}_{1.95}\text{Ti}_{0.05}\text{O}_6$  and  $0.05 \mu\text{C}/\text{cm}^2$  and  $10 \times 10^{+03} \text{V}/\text{cm}$  for  $\text{CoNb}_{1.9}\text{Ti}_{0.1}\text{O}_6$ .

**Chapter 7** provides the conclusions drawn from the investigation and presents the role of structural anisotropy or multiple cations doping to alter the relative ionic strength or ionicity of the M-O bond in the lattice to generate effective net polarization and long-range dipole-dipole interaction for the development high- $\kappa$  dielectric and relaxor type ferroelectrics. Insight for future investigation and a way forward to develop high- $\kappa$  dielectric and ferroelectrics.